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A study on resistive switching behaviors in rare earth oxide based MIM structure for memory application

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Chapter 1 Introduction

1.1 Background of this work:

Non-volatile memory technology plays a significant role in the market of electronics products. It is widely used in mobile phone, digital camera, portal storage devices, MP3 players and so on. Until now, Flash memory dominates the market of non-volatile memories, whose share of the market is above 90%. However, with the rapid scaling of microelectronics technology, Flash memory, which is based on the traditional floating gate concept, has encountered serious technical challenges due to its floating gate structure. Fig.1 schematically shows the principle of Flash memory: By storage different amount of electrons into the floating gate, the MOSFET devices shows different threshold voltage, which can represents different states. The problem of this structure comes from trade-offs between the high speed, low power operation and long time retention: high speed and low power requires a small capacitance between the floating gate and the channel, which long retention time requires a large capacitance.

Fig.1.1 Flash memory and its floating gate structure
Under this circumstance, several non-volatile memories based on different concepts have aroused extensive attention from both of industry field and academic field, such as Magnetoresistive RAM (MRAM), Phase change RAM (PRAM) and Resistive RAM (ReRAM). Graph 1.2 shows the comparison on several parameters of MRAM, PRAM, and ReRAM. Compared with other types, ReRAM shows relatively high speed, simple structure and the lowest power consumption.

<table>
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**Diagram 1.2 Comparison of ReRAM, Flash, PRAM and MRAM on speed, structure and power**
1.2 Introduction of ReRAM:

1.2.1 General introduction of ReRAM

Fig. 1.3 The basic principle of ReRAM

Resistive switching Random Access Memory (ReRAM) is a kind of memory that utilizing electric field induced resistance change in certain kinds of materials. It was proposed by Prof. Ignatiev from University of Houston in 2000, they developed a new method technology by utilizing the changeable resistance of Pr$_x$Ca$_{1-x}$MnO$_3$ under pulses with different polarity [1]. The basic working principle of ReRAM is shown in Fig. 1.3, the device is able to be abruptly switched between high resistance state (HRS) and low resistance state (LRS) by applying write voltage or erase voltage. HRS represents off state while LRS represents on state. In additional, the write voltage and the erase voltage do not need to continually turn on to keep the HRS or LRS. In other words, resistive switching is a kind of non-volatile memory effect.

Nowadays, ReRAM aroused more and more attention as a promising candidate for next generation of non-volatile memory. Here, we summary main advantages of ReRAM:

(1) ReRAM is completely compatible with current CMOS technology, which indicates that it is not necessary of great investment of new processes.
(2) ReRAM can be operated under very small power consumption. In the 0.18um process, the operation current can be controlled at the orders of $10^{-6}$A. 

(3) ReRAM have simple capacitor structure with two terminals, which is easy to be integrated to VLSI circuits.

However, ReRAM still facing several challenges to practical application. The key problem is the stability of the device performance, especially the endurance characteristics still requires further improvement. Another problem comes from the switching mechanism, which is still lack of quantified analysis. It is difficult to improve the device performance without a clear model.

1.2.2 Structures and Operation Method of ReRAM

![Fig. 1.4 The structure of single cell and arrays of ReRAM [2]](image)

The left figure in Fig. 1.4 shows the structure of a single ReRAM cells. It has a very simple capacitor-like structure, in which an insulating or semiconducting oxide is sandwiched between two metal electrodes. The right figure shows a possible
“cross-point” array for organizing ReRAM cells. Word and bit lines are used for selecting a memory cell and writing or reading data, respectively. Cross-point array is a two dimensional structure, which is relatively simple and requires less fabricating process. What’s more, cross-point array is able to achieve a very high storage density. The storage density is determined by the pinch of fabricating process. The definition of pinch of cross-points array is shown in Fig. 1.5.

Fig. 1.5 The definition of Pinch in cross-points array

Fig. 1.5 Two types of resistance switching behaviors [3]

The resistance switching behaviors of ReRAM devices can be divided to two basic types according to their I-V characteristics under sweeping voltage. It should be note
that these two I-V curves just show typical characteristics of the two different types, it can vary according to specific system. Besides, the dashed lines in the figure show the compliance current, which results in the difference between control voltage and real voltage in system. Fig. 1.5 (a) shows unipolar switching behaviors, in which the reset and set voltage is not dependent on the polarity of voltage. The set voltage is always higher than the reset voltage, and the reset current is always higher than the CC during set operation. Fig. 1.5 (b) shows bipolar switching behavior, in which the set and reset processes happen in the opposite voltage polarity. In some bipolar systems, CC is even not necessary.
1.2.3 Resistive switching mechanism in ReRAM

Although resistance switching behaviors have been found in a wide range of materials, and lots of models are proposed to explain those phenomenons, there are still lots of unclear places in the switching mechanisms. Generally, the resistance switching mechanism can be divided to three types in terms of position. They are interface effect, bulk effect and combination of bulk effect and interface effect. In this sub-section, we will introduce some switching mechanism at interface and bulk, respectively.

1.2.3.1 Interface effect

It has been certificated that in some ReRAM system, such as the contact of Pt electrode and SrTiO3[4], the change of schottky barrier should be responsible for the resistance switching phenomenon. The elimination and generation of oxygen vacancies leads to
the change of schottky barrier and thus change the tunneling current. For example, in
the case shown in Fig. 1.6, under the effect of negative voltage, the electrons combine
with the oxygen vacancies at the interface. This process results in a wider schottky
barrier and thus a smaller tunneling current, which corresponds to the HRS. On the
contrary, when the positive voltage is applied, opposite process happens and leads to the
LRS.

1.2.3.2 Bulk effect

1.7 Sketch of filamentary conduction in MIM structure

Bulk effect also play significant role in many kinds of oxides. Among several bulk
mechanisms, conductive filaments (CFs) are the most existed bulk effect that is
responsible for resistive switching. Formation of CFs can be classified as two types: 1)
CF formed by metal ions and 2) CF formed by oxygen vacancies. The first type relies
on an electrochemically active electrode metal. Fig. 1.8 hematically shows resistive
switching process in certain ReRAM device having Ag electrode. Ag atoms lose
electrons at anode and formed highly mobile Ag\(^+\) cations. Then it drifts under electric
field and get electrons at cathode to form Ag dendrites. Finally, the dendrites grow to
filaments that connect the electrodes, which set the device to ON state. Upon reversal of polarity of the applied voltage, and electrochemical dissolution of the filaments takes place, resetting the system into the OFF state. On the other hand, CF composed by oxygen vacancies also formed under similar electrochemical process. Under certain electric field, oxygen vacancies are created and forms filament under the electric field. And the filament is desolated under opposite electric field.

Fig. 1.8(a) Schematic illustration of resistive switching based on the motion of metal ions[5]

Fig. 1.8(b) Schematic illustration of resistive switching based on the motion of oxygen vacancies [6]
1.4 Parameters for evaluation of memory device performance:

In order to systematically evaluate the performance of ReRAM devices fabricated in this study, we use following indicator to depict and compare performance of devices.

1. Non-volatility: Volatility and Non-volatility are basic character of memory devices, which indicate that if devices are able to keep the programmed information after take off electrical powers. For example, SRAM and DRAM is volatile while Flash memory is non-volatile.

2. Operation voltage: Operation voltage depicts the necessary bias that makes the device works well. Large operation voltage not only puts negative influences on reliability, but also increases the complexity of external circuits. Incidentally, large operation voltage is one of weak points of Flash memory.

3. Memory window: Memory window represents the difference between different memory states. Its specific definition is different in different types of memory. For ReRAM, it is defined as the ratio of high resistance states to low resistance states. Window is directly related to the correction of data reading. A small or unstable window leads to read errs.

4. Retention: It is defined as the time that devices can keep programmed values. High quality non-volatile memories are able to keep values more than ten years at room temperature.

5. Multilevel storage: Multilevel storage indicates the devices have more than two states,
which improve the storage density. In other words, compared to device with binary states, a multilevel device can represent more than two values.

6. **Endurance**: It is the number that how many times the device can be wrote and erased before failure. For non-volatile memory, $10^6$ times switching is an ideal value.

7. **Write/Erase speed (W/E speed)**: The minimum time required by devices for writing and erasing. The W/E time of Flash memory is at the order of $10^{-6}$ s to $10^{-3}$ s. On the other hand, the W/E time of ReRAM can achieve the orders of 10-100ns.

8. **Scaling**: The potential of the device for scaling without sacrificing its memory performance.
1.5 Purpose of this work

Resistive random access memory (ReRAM), which utilizing resistive switching of certain materials, has attracted much attention as non-volatile memory in the near future because of its high speed, low consumption and great potential for scalability [3]. Resistive switching behaviors in various insulating oxides, ranging from simple binary oxides such as NiO$_2$ [6] and TiO$_2$ [7] to complex oxides such as perovskites [8] and chalcogenides [9], has been reported. Until now, relatively few studies on the resistive switching behaviors of cerium oxides has been reported. Among various insulating oxides, cerium oxides have strong potential for ReRAM application for the following reasons: 1) cerium oxides have high dielectric constant and moderate band gap, 2) cerium atom exhibits both +3 and +4 oxidation states in cerium oxides, which is suitable for valency change switching processes[5], 3) oxygen ions/vacancies, which play a important role in resistive switching[2], produce high conductivity in CeO$_2$ due to its fluorite structure, and 4) cerium oxide is easy to react with Si to form silicate [8]. The last reason provides us a method to modify the structure and concentration of vacancy in cerium oxide film by forming cerium silicate. In this paper, cerium oxides based ReRAM devices with silicon buffer layer were fabricated by utilizing W and TiN as electrodes. The resistive switching behaviors of the devices and the influence of the Si buffer layer on the resistive switching were investigated in detail.
Reference:


Chapter 2 Fabrication and Characterization

Method

In this chapter, all of the fabricating and characterization methods that have been utilized in studies are introduced. The principles of processes and equipments are simply discussed and the experiment conditions are also listed.

2.1 Fabrication Method

2.1.1 Si surface cleaning

Si wafer cleaning is critically important in the era of VLSI and ULSI technology. Over 50% of yield losses in integrated circuit fabrication are generally accepted to be due to microcontamination[1], such as metal contamination, organic contamination, ionic contamination, and etc. The cleaning process for wafers used in this studies starts from rinsing Si wafers with running DI (de-ionized) water. The DI water used in this work, incidentally, has a resistance high than 18.2 MΩcm, while the theoretical resistance of pure water is 18.25 MΩcm at 25 °C. Then the samples were treated by sulfuric-peroxide mixture (SPM), which is composed by H₂O₂ and H₂SO₄ (H₂O₂: H₂SO₄ = 1:4), for 10min at 120°C. During this process, existed organic and metallic contamination was separated from surfaces of Si wafers because of chemical oxidation of the surfaces. After that, Si wafers was rinsed with running DI water again. In order to remove the chemical oxide formed in SPM process, the wafers are further treat by diluted HF (1%HF) for 1min. Finally, wafers are cleaned again by running DI wafer before next
fabrication process.

2.1.2 Thermal Oxidation Process

Thermal oxidation is one of key processes in VLSI technology. In the case of silicon dioxide, it can be classified to two types: wet oxidation (\(\text{Si} + 2\text{H}_2\text{O} = \text{SiO}_2 + 2\text{H}_2, \quad 800^\circ\text{C}-1100^\circ\text{C}\)) and dry oxidation (\(\text{Si} + \text{O}_2 = 2\text{SiO}_2, \quad 800^\circ\text{C}-1100^\circ\text{C}\)). Wet oxidation is suitable for growth of thick silicon dioxide because of its relatively high growth rate, while dry oxidation is rather slow but easily controllable. In this work, a layer of SiO\(_2\) (about 200nm) is grown on a Si wafer by dry oxidation in an oxidation furnace, whose structure is shown in Fig.X. One of critical challenges in thermal oxidation is creation a gas ambient with precise, constant and uniform temperature. By utilizing three heaters in different part of the quartz tube, the temperature becomes more uniform in the whole tube. In addition, only a small localized region in the middle of tube is used for thermal oxidation, which further improves the uniformity of the temperature and the mass flow. The samples are set on a small quartz boat and slowly send into the middle of the tube by a quartz rod. The mass flow of O\(_2\) is controlled to be 1L/min and the oxidation is continued for 2 hours at 1000\(^\circ\text{C}\).
Fig. 2.1 (a) Schematic illustration of the oxidation furnace

Fig. 2.1 (b) Photo of oxidation furnace
2.1.3 Photolithograph

In order to fabricate desired device used for this study, the photolithograph process is utilized for two times: One time for patterning the window between the bottom electrode and the substrate, and the other for patterning the top electrode layer to square-shape with different area. The process flow of the photolithography has been carried out during this research is shown in Fig. A. First of all, a uniform thin layer of positive photoresist was formed by high speed spin coating. Then the wafer was heated to 115oC to drive off excess moisture in the photoresist, which is so called pre-baking process. Next, the wafer coated by photoresist was aligned and exposed through e-beam patterned hard-mask with high-intensity ultraviolet (UV) light at 405 nm wave length. MJB3 of Karl Suss contact-type mask aligner [3] was used in this process. The exposure duration was set to 4.8 sec. After that, developing process is performed by a specified developer called NMD-3 (Tokyo Ohka Co. Ltd.) for 2min. Finally, the resulting sample was treated by post-baking process to solidify the remaining photoresist for the following wet chemical etching and plasma etching.
2.1.4 Etching process

Reactive ion etching (RIE) and buffered HF (BHF) etching are both adopted for etching process in this work. RIE is a widely used dry etching technology, whose basic working principle is shown in Fig. 2.3. Etching gas is introduced into a low pressure chamber and then plasma is produce by a strong RF (radio frequency) electromagnetic field. In each cycle of the filed, electrons are accelerated to high velocity by the electric field because of its neglectable mass, while the velocity of positive ions are very slow as their far larger mass. As a result, by applying a RF electric field, electrons strike both of the upper plate and the bottom plate of the chamber, while the positive ions are concentrated between the plates. The upper plated is grounded and stroked electrons flow away. However, electrons build up negative charges on the wafers because of their insulating characteristic. Consequently, a strong local electric filed is established.
between the positive etching ions and the negative wafers. The ions bombard the sample and chemically react with certain kind of material, which results in etching selected part of the sample. Furthermore, physical sputtering of the sample surface also happens during this process. Because of the vertical electric field, RIE shows anisotropic etching profile. In this work, SF$_6$ was used to etching the W electrode layer and O$_2$ was used to remove the photoresist from the etched wafer. On the other hand, BHF is also adopted for wet etching of silicon dioxide over silicon wafers, which is an isotropic etching process. BHF is composed by HF, NH$_4$F and H$_2$O, and it can etching silicon dioxide in a controlled speed (100nm/min for BHF used in this work),

![Fig. 2.3 Schematic illustration of Reactive ion etching](image)

### 2.1.4 E-beam evaporation and RF magnetron sputtering:

E-beam evaporation and RF magnetron sputtering are used extensively in the semiconductor industry to deposit thin films of a large amount of materials in VLSI technology. E-beam evaporation is employed for depositing CeO$_2$ thin film in this work. Deposition is carried out at chamber in ultra high vacuum state (10$^{-8}$~10$^{-9}$ Torr). The sample is heated to 300 °C and rotated at a constant speed in advance. Electron beam, which is accelerated by a 5 KV electric field, is generated and bombard to the source
under the control of a magnetic sweep controller. Consequently, electrons flow though
the source and the molecules evaporate because of Joule heat. The growth of the thin
film is observed by a thickness monitor and the growth rate in this work is controlled at
0.005 A/s to 0.008 A/s to ensure a relatively high film quality.

![Fig. 2.4 Schematically illustration of E-beam Chamber](image)

On the other hand, RF magnetron sputtering is also adopted for depositing the electrode
(TiN and W) and the Si buffer layer. The sputtering process is schematically shown in
the Fig.2.5, Argon gas is released into the chamber and then Ar plasma was generated
under strong RF electron field and magnetic field. By magnetic field, ions follow a
helical path, which is able to causes more ionizing collisions and improve deposition
rates. After the generation of plasma, Ar⁺ ions were accelerated to the target and sputter
the target atoms to the substrate by ion bombardment. During the process of film growth,
RF power was supplied to avoid the charge build-up on insulation targets. Besides, The
TiN film in this work was produced by reactive sputtering: N₂ gas was also introduced
into the chamber in this case and reacts with Ti target to form Nitride films.
Fig. 2.5 (a) Schematic illustration of RF magnetron sputtering [2]

Fig. 2.5 (b) Photo of the sputtering systems
2.1.5 Rapid Thermal Annealing

Rapid thermal annealing (RTA) stands for a kind of thermal treatment that heats samples to high temperatures in a short time, which in the orders of seconds. And usually, a slow cooling down process is followed for protect the sample from break due to abrupt temperature change. It can be used for re-crystallization, activation of dopants, diffusion of ions between different film, recovery of damage and defects, and formation of new chemical substance. Besides, the high temperature ramp rate is usually achieved by lamp or laser, and the annealing ambient is usually N\textsubscript{2} or forming gas (N\textsubscript{2} + H\textsubscript{2}). In this work, the fabricated samples are treated by rapid thermal annealing at the last stage for recovery of defects and formation of silicate by CeO\textsubscript{2} and Si. The equipment of QHC-P610CP (ULVAC RIKO Co. Ltd) is utilized, whose structure is shown in Fig. 2.6. Samples are heated by an infrared lamp system and cooling down by flowing water. The annealing condition is 400\textdegree{}C for 30s in N\textsubscript{2} gas ambient (1 L/min). All the samples were taken out of the chamber under 100\textdegree{}C.
Fig. 2.6 (a) Schematic illustration of the infrared annealing furnace

Fig. 2.6(b) Photo of the infrared lamp system
2.1.6 Thermal evaporation for Al backside contact

Aluminum (Al) is evaporated to the backside of the samples in this work for the better contact of the substrates and the ground during measurement, which is achieved by using bell-jar type thermal evaporation as illustrates in Fig. 2.7. This system utilized a turbo molecular pump (TMP) to achieve background pressure up to 1.0x10^{-5} Pa prior to Al evaporation. Filament which used to hold Al wires is made of tungsten (W). Purity of W filament and Al source is 99.999%. Methanol and acetone were used to clean the W filament and Al wires prior to every evaporation process. Chamber pressure during evaporation is 2x10^{-5} - 5x10^{-5} Pa. A DC current about 60A was used to evaporate Al.

Fig. 2.7 Schematic illustration of the structure of Bell Jar [4]
2.2 Characterization method

2.2.1 Current-voltage measurement

I-V measurements were done on HP4156A semiconductor parameter analyzer. In order to tracing the resistance switching behaviors, a sweep voltage ranging for -10V to 10V, whose step is 0.05 V, is used. Besides, different compliance current ranging from 1mA to 5mA is also set to protect the device from break down. In addition, in order to investigate the conduction mechanism of different conduction state, I-V measurement in different temperature is also carried out.

Reference


Chapter 3: Resistive switching behaviors in the RRAM device having W/CeO$_2$/Si/TiN Structure

3.1 Device design and fabrication
The structure of the device and the measurement circuit are schematically shown in Fig. 3.1.2, while fabrication process flow is shown in Fig. 3.2. Highly doped Si wafer was used as substrate for better contact between the bottom electrode and the cathode. A 200-nm-thick SiO$_2$ insulator layer was grown on the substrate by thermal oxidation to protect the device from leakage current, and contact windows though the SiO$_2$ layer were patterned for the connection between the bottom electrode (BE) and the substrate. Then, a 15-nm-thick TiN BE layer and a 1-nm- or 2-nm-thick Si buffer layer were successively formed by RF sputtering. Next, a 20-nm-thick CeO$_2$ layer was deposited by electron beam evaporation as the resistive switching layer, then a 50-nm-thick W top electrode (TE) layer was in-situ deposited by sputtering and was patterned to form electrodes having area of 20 \( \mu \text{m} \times \text{square}. \) Finally, Al was evaporated as a back contact.
followed by rapid thermal annealing (RTA) in N$_2$ ambient for 30 s at 400°C. Furthermore, the device without Si buffer layer was also fabricated to evaluate the influence of Si buffer layer on switching characteristics.

Fig. 3.1.2 W/CeO$_2$/Si/TiN device fabrication process flow

3.2 Resistive switching behaviors
Fig. 3.2.1 I-V curve of W/CeO$_2$(20 nm)/Si(1 nm)/TiN for the following range of bias voltages in each cycle: ① 0 V to 3 V, ② 3 V to 0 V, ③ 0 V to -3 V and ④ -3 V to 0 V. A forming process is necessary in the first cycle.

The typical current-voltage curve measured at room temperature is shown in Fig. 3.2.1. The bias potential was applied to the W TE, and the TiN BE was grounded. A forming process of fresh W/CeO$_2$/Si/TiN device at the first cycle requires low voltage because Si buffer layer is introduced. The devices can be operated in the range of voltage from -3 V to 3 V with compliance current (CC) of 1 mA. According to this figure, bipolar resistive switching from high resistance state (HRS) to low resistance state (LRS) under positive bias is observed, while it returns to HRS under negative voltage. The endurance characteristic and retention characteristic are illustrated in Fig. 3.2.2. The device is able to keep a nearly 30 times big window even after 60 cycles and also keep the window for $10^5$ s. The unique feature of this device does not require high voltage in the forming process. As a result, this process not only protects devices from irreversible breakdown,
but also simplifies the external circuits of ReRAM.

Fig. 3.2.2 (a) Endurance characteristics of W/CeO$_2$(20 nm)/Si(1 nm)/TiN device

Fig. 3.2.2 (b) retention characteristics of W/CeO$_2$(20 nm)/Si(1 nm)/TiN device
Figure 3.2.3(a) shows the I-V relationships at HRS and LRS, whose conduction mechanisms are clearly different to each other. Current and voltage follow linear relation, in other words ohmic relation, at LRS and implies the formation of conductive filament (CF). Figure 3.2.3(b) shows the temperature dependence of HRS and LRS. With increase in bath temperature, resistance of HRS decreases like a semiconductor. On the other hand, with the increase in the bath temperature resistance of LRS gradually increase like a conductor.

Fig. 3.2.3 (a) I-V curve of set process in linear scale
Fig. 3.2.3 (b) Temperature dependence of HRS and LRS of W/CeO$_2$(20 nm)/Si(1 nm)/TiN device

Fig. 3.2.4 The dependence of HRS and LRS on the electrode area

Fig. 3.2.4 shows the dependence of HRS and LRS on the electrode area. The values showed in the graph are the average value of ten times switching. It shows a very import
tendency that the LRS is nearly independent with electrode area while the HRS increase as electrode area decrease. In addition, the memory window also get larger in smaller area. This phenomenon further demonstrates the conductive filament mechanism in the resistance switching in cerium oxide: Due to conductive filament is only formed at a localized places, it largely depend on the operation voltage and the compliance current instead of electrode area. On the other hand, HRS, which corresponds to the state that conductive filaments rapture, is largely dependent on electrode areas and smaller areas leads to higher resistance. However, it is also can be observed that the distribution of LRS of 10um*10um devices is larger than the one of 20um*20um devices, which possibly can be attributed to the degraded uniformity at smaller area. The area dependence of the W/CeO$_2$/Si/TiN not only further certificate the conductive filament mechanism is responsible for the resistance switching, but also show that the device has strong potential for further scaling since the enlarged window as scaling.

In summary, The ReRAM device with W/CeO$_2$/Si/TiN structure shows stable resistive switching behaviors. It is able to keep a 30 times big window even after 60 switching cycles and its retention time is longer than $10^5$ s. The conduction mechanism, temperature dependency and area dependency indicates that the resistance switching can be attributed to the conductive filament mechanism[1]. In addition, the device shows strong potentials for further scalability.
3.3 The effect of Si buffer layer

Fig. 3.3.1 Typical I-V curve of W/CeO$_2$(20 nm)/TiN device. A large voltage is necessary for the forming process at the first cycle. Typical switching behavior of the device with 1-nm-thick Si layer is also shown by the grey lines.

Fig. 3.3.2 Endurance characteristic of the W/CeO$_2$(20 nm)/TiN device (filled circle) and
that of the device with 1-nm-thick Si layer is also shown (open circle).

In order to investigate the influence of Si buffer layer, the device without Si buffer layer, which has W/CeO$_2$/TiN device structure and was treated in same annealing condition, was also investigated. As shown in Fig. 3.3.1, the device without Si buffer layer requires a high voltage for the forming process. The device without Si buffer layers also requires a high CC compared with the device with Si buffer layer. Figure 3.3.2 shows the endurance characteristic of the device, and the window is relatively small (about 10) and degrades quickly in accordance with switching is repeated. Furthermore, the resistance at HRS of device without Si buffer layer is far lower than that of the device with Si buffer layer. This implies that in most part of cerium oxide the irreversible breakdown of oxide film occurs in the device without Si buffer layer. Fig. 3.3.3 shows the distribution of set voltage of the ReRAM devices with and without Si buffer layer when the Vstop is fixed at -3V. It shows that the set voltage of the device with Si buffer layer

![Bar chart showing the distribution of set voltage of devices with and without Si buffer layer.](chart_image)

Fig. 3.3.3 The distribution of set voltage of devices with and without Si buffer layer, and the Vstop is fixed at -3V.
get more concentrated than those without Si buffer layer. This result indicates that the high forming voltage and larger compliance current also decrease the stability of the ReRAM device.

Simply puts, by inserting a Si buffer layer, the resistive switching is easier to be triggered at lower voltage with lower CC. This protects the device from hard breakdown, and thus overall improves the device performance in terms of enlarged memory window, better endurance characteristic, lower power consumption and more stable distribution of operation parameters.
3.4 Proposed model for explanation of the effect of Si buffer layer

Fig. 3.4.1 The forming process and typical switching cycle of the device without Si

Fig. 3.4.2 The forming process of the device with Si buffer layer
Although the conductive filaments (CFs) was not directly observed in this work, we apply the model based on the formation of CF to explain the experimental results by considering ohmic relation in LRS and the forcing process. The model is schematically illustrated in Fig. 6. Figure 6(a) shows the normal forming process of device without Si buffer layer. There is not any pre-existed CF in the thin film at initial state, and thus the formation of CF at fresh devices requires a large voltage. After the formation of CF, the large voltage is not necessary for the set and reset processes are just the partially rapture and formation of the CF. Fig. 6(b) shows the formation of CF in the devices with silicon buffer layer. After suitable thermal treatment, for example, in N₂ at 400°C for 30s, oxygen vacancies can be introduced near the interface by the reaction between cerium oxide and silicon (CeO₂ + Si = CeSiO₄ + 2V₉́ or Ce₂O₃ + Si = Ce₂SiO₅ + 2V₉́). As a result, some of oxygen vacancies diffuse along the GBs and thus partial part of CF has been formed at the initial state, which allows the forming process to be finished at lower voltage. Consequently, hard breakdown of the oxide film caused by high voltage is avoided and the device is able to work at low compliance current.
3.5 Thickness effect of the silicon buffer layer

Fig. 3.5.1 Typical forming process and resistive switching of the device with 1-nm- and 2-nm-thick Si buffer layer

Fig. 3.5.2 Resistive switching vs switching cycles of devices with 1-nm- (filled square) and 2-nm- (open circle) thick Si buffer layer
The influence of Si buffer-layer-thickness on the resistive switching of W/CeO$_2$ (20 nm)/Si(2nm)/TiN device, which was treated by same annealing condition, was also studied, whose typical I-V curve and C-V characteristics and the change in resistance as a function of switching cycles are shown in Figs. 3.5.1 and 3.5.2, respectively, where the corresponding values of the device with 1-nm-thick Si buffer layer are also shown for comparison. As shown in Fig. 8, the window of the device with 2-nm-thick Si buffer layer is smaller than those with 1-nm-thick Si buffer layer. Furthermore, two kinds of devices have similar LRS and the difference comes from HRS. Using the aforementioned model, the thickness effect of the Si buffer layer can be explained by the following mechanism Although thicker Si layer results in the introduction of larger amount of oxygen vacancies, the formation of CFs is largely determined by set voltage and CC. Because the devices with 1-nm- and 2-nm-thick Si buffer layer are operated at close voltage with the same CC, similar CFs are formed to have similar LRS in these devices. However, larger amount of oxygen vacancies make the rapture of CFs become more difficult. As a result, the device with 2nm Si layer is not able to reset to a HRS as high as the one of the device with 1nm, which also can be observed in Fig.3.5.1. The influence of the Si buffer-layer-thickness on the resistive switching indicates that in order to achieve the best performance of the devices, the thickness of Si buffer layer should be optimized.
3.6 Conclusion

In summary, W/CeO$_2$/Si/TiN ReRAM device shows bipolar resistance switching behaviors. It is able to keep a nearly 30 times big window even after 60 cycles and also keep the window for $10^5$ s. The conduction mechanism of LRS, the temperature dependency of HRS and LRS, and also the area dependency demonstrates that the conductive filament mechanism should be responsible for the resistance switching. It was also find that the Si buffer layer play a significant role in such resistance switching behaviors, which effectively increase the memory performance of the devices. The Si buffer layer not only decreases the voltage for the forming process, but also enlarges the window, improves the endurance characteristic and lowers power consumption. Furthermore, a model based on conductive filament mechanism was found to explain the physical effect of the silicon buffer layer, which is attributed to the introduction of additional oxygen vacancies and thus help resistance switches more easily. Finally, the thickness effect of the Si buffer layer is investigated. It is conclude that thicker Si buffer layer leads to larger amount of oxygen vacancies and the thickness of Si buffer layer should be should be optimized to obtain the best performance of the device.

Reference

Chapter 4: Modeling of the resistance switching behaviors in W/CeO$_2$/Si/TiN device

4.1 The influence of Vstop on the resistance switching

Fig. 4.1.1 The influence of Vstop on the resistive switching behaviors of W/CeO$_2$/Si/TiN

Fig. 4.1.2 The reset processes fails at Vstop=3.8V and Vstop=1.4V
In the reset process of the resistive switching behaviors of W/CeO$_2$/Si/TiN ReRAM devices, it can be clearly observed that there is not any sharp reduce in the reset current. Insteadly, the reset process in this process seems a gradual change process and relates to the maximum of sweeping negative voltage. Therefore, this work investigates the influence of Vstop on the resistive switching behaviors of the device. As shown in Fig. 4.1.1, as the Vstop shift to negative direction, the amplitudes of resistive switch become larger. It indicates that the reset process is not an abrupt process in this device. In addition, a small “hub” at nearly 1.4 V can be observed in the reset processes with different Vstop. The small hub is actually a threshold voltage for reset process, the device is not able to reset when Vstop is smaller than 1.4 V (Fig. 4.1.2). Such a threshold voltage probably comes from localization effect of the electrode [1]. In the set process, oxygen vacancies are formed at anode and lattice oxygen ions are tied by W electrode because W is an oxidable electrode. And in the reset process, when the oxygen ions tied by electrode need to recombine with oxygen vacancies, a threshold voltage is required to provide oxygen ions enough energy to jump over the interface barrier. Besides, It is also shown in Fig. 4.1.2 that the Vstop also can not increase arbitrarily, the break down in negative position will happen when it larger then 3.8V.

![Fig.4.1.3 The activity energy of oxygen ions](image)
Fig. 4.1.4 Resistive switching of the device during continually change Vstop from 3.2 V to 1.6 V at a step of 0.2 V every 5 times switching cycles.

Fig. 4.1.4 shows the HRS and LRS as continually change Vstop from 3.2 V to 1.6V at a step of 0.2 V every 5 times switching cycles. During the change of Vstop, HRS changes sharply while LRS nearly keep stable. It indicates that the reset process of W/CeO$_2$/Si/TiN is continually carried out during |Vstop| increase. Furthermore, although the HRS is not stable, there are still two distinguishable windows can be observe, as shown by the dashed line in the figure. Such phenomenon shows the potential of W/CeO$_2$/Si/TiN devices for multi-level storage by controlling Vstop.
4.2 Modeling the ratio of fractured filaments on reset process

As shown by W/CeO$_2$/Si/TiN, it evidences the rapture of multi-filaments. Fig. 4.1.4 shows the model based on parallel connection of filament, in which different resistance states correspond to the different situation of connection of each filament. In our devices, when the Vstop exceed the switching barrier, the reset process is triggered and the filaments begin to rapture. In the following work, we will clear out the relationship between the Vstop and the number of filaments that rapture at that Vstop.

![Resistive switching model for multi-filaments](image)

Fig.4.1.4 Resistive switching model for multi-filaments
fractured filaments in reset process (%) vs. $|V_{\text{stop}}| (V)$.
Chapter 5 Conclusion

In this study, a new method for improving the cerium based ReRAM devices by utilizing Si buffer layer is firstly proposed in consideration of the special characteristics of CeO$_2$ and Si. Then the ReRAM device having W/CeO$_2$/Si/TiN structure is fabricated and characterized in detail. Next, by comparison with the device without Si layer (W/CeO$_2$/TiN), the effect of Si buffer layer is extracted and explained by an model based on conductive filament mechanism. In addition, the thickness effect of Si buffer layer is also clarified by experiment results. Lastly, the influence of voltage that applied during set and reset process on resistive switching behavior of the device is modeled, which provides a strategy for further improve the device performance. The conclusions in this work is summarized as following:

1) The ReRAM device having W/CeO$_2$/Si/TiN (and also W/CeO$_2$/TiN) structure show bipolar resistive switching behaviors, which can be attributed to conductive filament mechanism in consideration the conductive mechanism and area dependence of different resistive states.

2) The W/CeO$_2$ (20nm)/Si(1nm)/TiN device has a nearly 30 times big window even during 60 cycles and also can keep the window longer than $10^5$ s, which is far more better than that of the device without Si. Besides, the device with Si doesn’t require a high voltage in forming process and also can work under lower compliance current.

3) The effect of Si buffer layer can be explained by following mechanism: The Si buffer layer reacts with the lattice oxygen ions at the surface of CeO$_2$ and thus
introduction oxygen vacancies into the interface, which eventually leads to an easier forming process.

4) Thicker Si buffer layer leads to larger amount of oxygen vacancies and the thickness of Si buffer layer should be optimized to obtain the best performance of the device.

5) The stage-like HRS caused by continuing change of Vstop evident the existence of multi-filaments during restive switching process. And rupture of the filaments is directly related to Vstop.
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